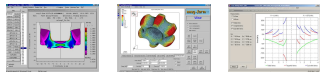


## ***“Various Approaches to deal with Microstructure in Simulation of Bulk Metal Forming”***

*Dr. Gerhard H. Arfmann, Dr. Michael Twickler  
CPM GmbH, Herzogenrath, Germany*



## ***“Various Approaches to deal with Microstructure in Simulation of Bulk Metal Forming”***

*Why to determine the microstructure?*

*The most simple Model: “Pause Time = Recrystallization“*

*The typical Model: “Dynamic/Static with grain growth“*

*The complex model: “Dislocation Based Model“*

*Acknowledgement*



## Why to determine the microstructure?

### Motivation:

- most realistic modeling of the forming processes in the Simulation (Material flow, Strain hardening, etc.)
- prove of reaching a certain specified grain structure (average grain size, max permitted grain size , etc.)
- provide the necessary grain structure information for subsequent further treatment (heat treatment or others)
- derivation of product properties
- information to support further experimental investigations (i.e. orientation of flow lines for ultra sonic testing)
- etc.



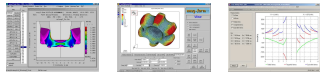
## The most simple Model: "Pause Time = Recrystallization"

### Basic approach:

- sufficiently high temperatures and a least energy brought in by forming result in recrystallization in the part (dynamic, static)
- the recrystallization reduces the deformation that is already in the part
- the local temperature and strain are always sufficient to enable recrystallization
- the pause time (transport of the part) is always sufficient to allow for complete recrystallization
- complete recrystallization is identical with complete reset of any deformation that is already in the part
- dynamic recrystallization during the forming will have no effect to the material properties nor to the material flow

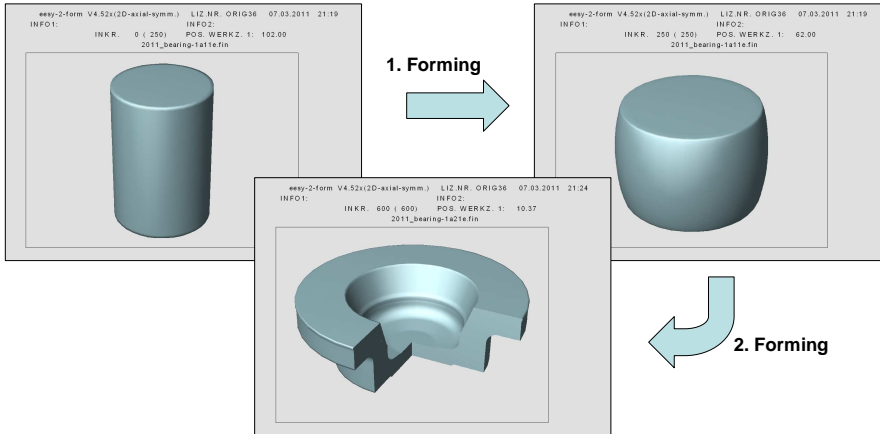


End of forming = Pause time = Complete recrystallization

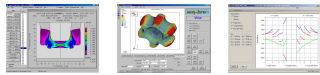


## The most simple Model: "Pause Time = Recrystallization"

Example: Bearing Shell, 2-Step Forming at 1150°C

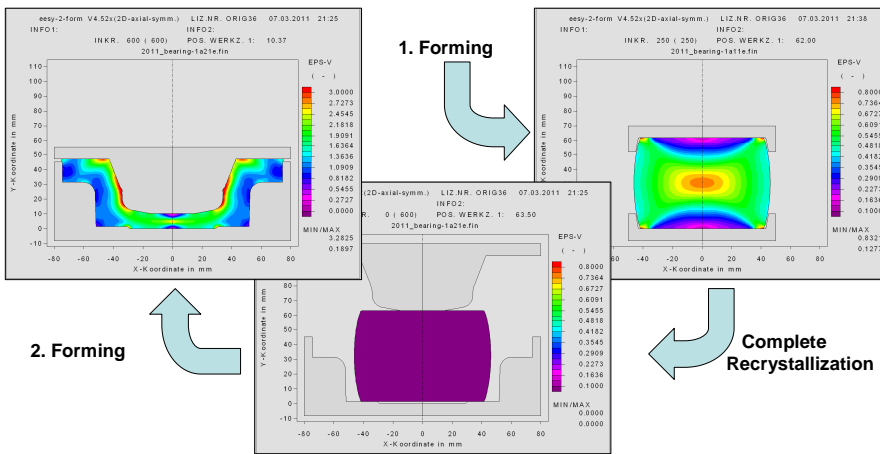


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## The most simple Model: "Pause Time = Recrystallization"

Example: Bearing Shell, 2-Step Forming at 1150°C



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## The most simple Model: "Pause Time = Recrystallization"

### Results:

- in a multi step forming process each step starts with completely recrystallized material behavior (there is no forming history)
  - ==> the resulting strain values often show a realistic picture of the real material situation
  - ==> a basic interpretation of the actual deformation regarding the limit of formability can be performed
  - ==> the calculated forming and tool loads are not influence by an accumulation of strain from previous operations
- Information about the state of recrystallization of the material: **not available**
- evaluable information about grain size or grain size distribution: **not available**
- information about grain growth: **not available**
- further microstructure relevant results: **not available**
- influence on the used material model: **none, Yield stress – strain curve remains unchanged**



## The most simple Model: "Pause Time = Recrystallization"

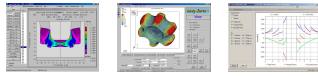
### Model idea (1):

- a model of the dynamic recrystallization can be formulated like

$$\text{dynamisch} \left\{ \begin{array}{l} D_{\text{dyn.rek.}} = f(D_0, \epsilon, \delta\epsilon/\delta t, T, \text{etc.}) \Rightarrow \text{dyn. recryst. grain size} \\ F_{\text{dyn.rek.}} = f(\text{several process parameters}) \Rightarrow \text{dyn. recryst. fraction} \\ \text{Relevance: during respectively directly after the forming} \end{array} \right.$$

- a model of the static recrystallization can be formulated like

$$\text{statisch} \left\{ \begin{array}{l} D_{\text{stat.rek.}} = f(D_0, \epsilon, T, t, \text{etc.}) \Rightarrow \text{stat. recryst. grain size} \\ F_{\text{stat.rek.}} = f(\text{several process parameters}) \Rightarrow \text{stat. recryst. fraction} \\ \text{Relevance: during pause time between respectively hold time after previous deformation} \end{array} \right.$$



## The most simple Model: "Pause Time = Recrystallization"

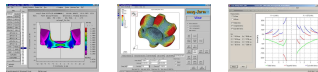
### Model idea (2):

- a model for grain growth can be formulated like

Wachstum {  $\Delta D_{xx,rek.} = f(D_{xx,rek.}, T, t, \text{etc.})$  with  $xx. = \text{stat. OR dyn.}$   
 => growth of the xx. recrystallized grain  
 Relevance: for  $\Delta D_{dyn,rek.}$ : after the end of the dynamic recrystallization  
 for  $\Delta D_{stat,rek.}$ : after complete static recrystallization of the leftover grainstructure

- it is possible to find reasonable rules to allow at a certain step of time to merge the separate grain fractions like

Mischung {  $D_{0-neu} = f(D_0, D_{dyn,rek.}, D_{stat,rek.}, \Delta D_{dyn,rek.}, \Delta D_{stat,rek.}, F_{dyn,rek.}, F_{stat,rek.})$   
 => new average of the initial grain size



## The typical Model: "Dynamic/Static with grain growth"

### Model idea (3):

- the reduction of the amount of strain in the part is equivalent to the sum of the recrystallized fractions (considered locally) like

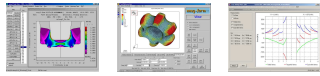
Strain Softening {  $\epsilon_{v-neu} = f(\epsilon_v, F_{dyn,rek.}, F_{stat,rek.})$  => new reduced strain  
 Relevance: after completion of the dynamic recrystallization and during the static recrystallization

- the complex interaction between the several mechanism of microstructure change can be formulated and coded. Normally there is no strict sequence how the various models interact during a process. Knowledge is needed about:

- interaction and dependencies of the various models
- Start/Stop-conditions for the models

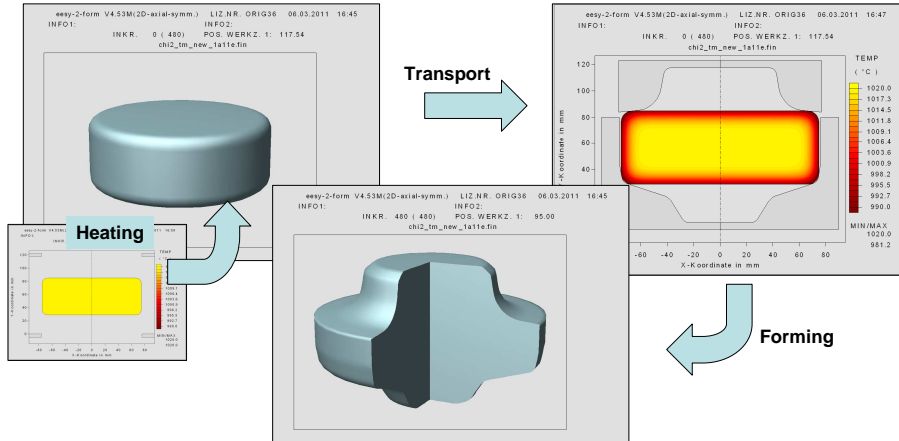


Dynamic/Static Model for Forming and Hold Times

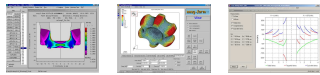


*The typical Model: "Dynamic/Static with grain growth"*

Example: Turbine disc made from Inconel 718, one step forming at 1020°C

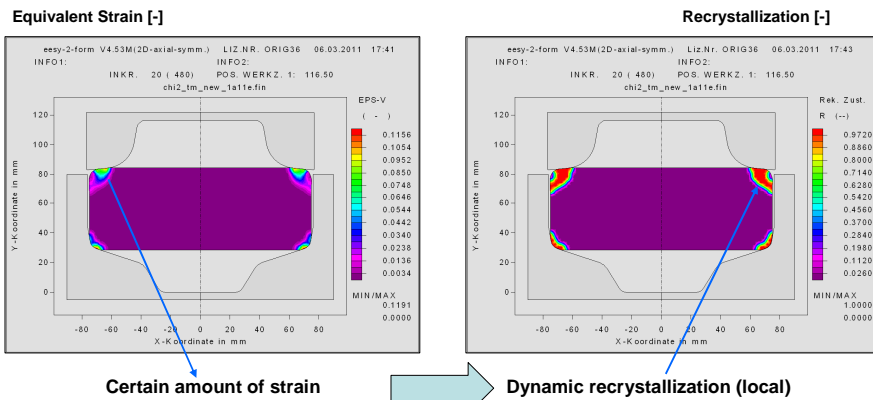


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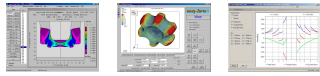


*The typical model: "Dynamic/Static incl. Grain Growth"*

Example: Turbine disc made from Inconel 718, one step forming at 1020°C  
- results after 5% press movement



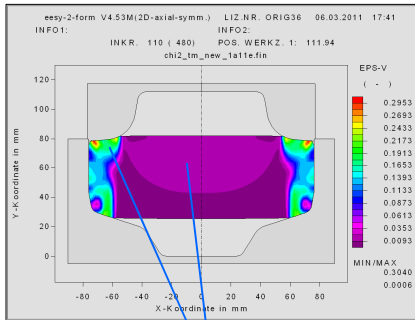
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*The typical model: "Dynamic/Static incl. Grain Growth"*

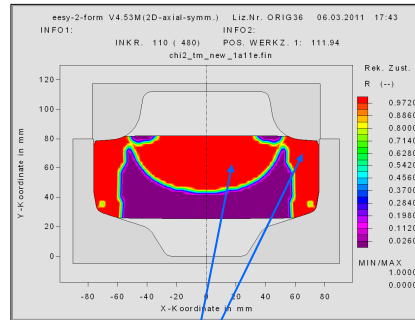
**Example: Turbine disc made from Inconel 718, one step forming at 1020°C  
- results after 25% press movement**

**Equivalent Strain [-]**



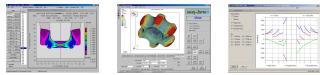
**Certain amount of strain**

**Recrystallization [-]**



**Dynamic recrystallization (local)**

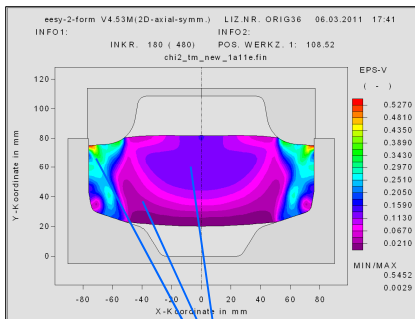
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*The typical model: "Dynamic/Static incl. Grain Growth"*

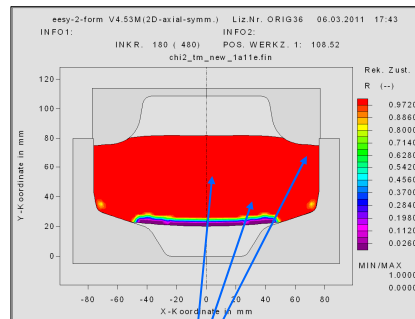
**Example: Turbine disc made from Inconel 718, one step forming at 1020°C  
- results after 40% press movement**

**Equivalent Strain [-]**



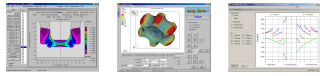
**Certain amount of strain**

**Recrystallization [-]**



**Dynamic recrystallization (local)**

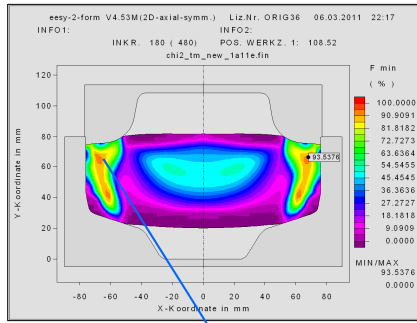
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*The typical model: "Dynamic/Static incl. Grain Growth"*

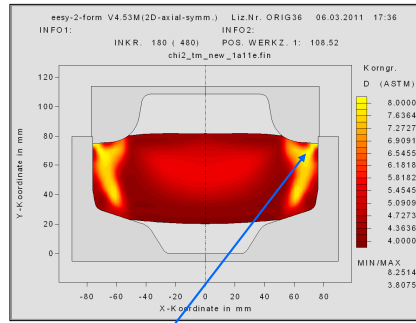
Example: Turbine disc made from Inconel 718, one step forming at 1020°C  
- results after 40% press movement

Recrystallized Fraction [%]



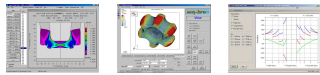
High recrystallized fraction

Grain Size [ASTM]



Fine (dyn. recrystallized) grain

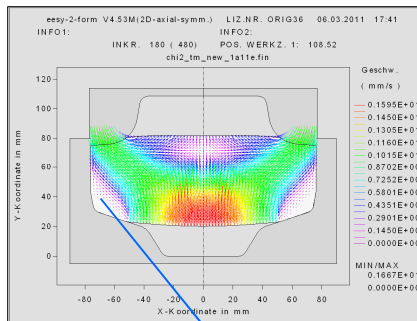
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*The typical model: "Dynamic/Static incl. Grain Growth"*

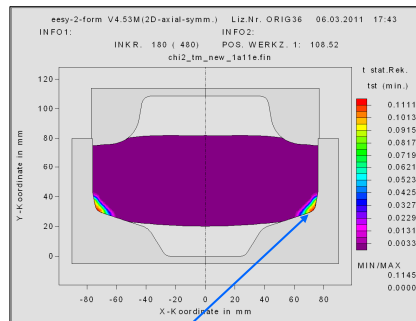
Example: Turbine disc made from Inconel 718, one step forming at 1020°C  
- results after 40% press movement

Material Flow [mm/sec]



„Dead“ zone, - actually no forming

Duration of Static Recrystallization [min]



stat. recrystallization after forming

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## The typical model: "Dynamic/Static incl. Grain Growth"

### Results:

- resulting strain represents the actual deformation and hardening situation (locale distribution)  
=> realistic calculation and evaluation of several variables like material stress, deformability, forming loads, tool loads
- state of recrystallization: available, split in dynamic and static fraction
- grain size and grain size distribution : available, as global value and in it's dynamic and static fractions as well
- Indication of grain growth : available, quantitative value
- further microstructure spec. results : available, i.e. recrystallization time, recrystallization stop, timing of growth, etc.
- all grain characteristics as local distribution in the part: available
- sustainable effects on the material model used : non, yield stress-strain curve remains unchanged



## The complex model: "Dislocation Based Model"

### Model idea (1):

- strain  $\epsilon$  as not path independent value is no longer used as state value for the constitutive law to describe the mechanical behavior  
i.e.

$$\sigma = f(\delta\epsilon/\delta t, T, \epsilon) \quad \text{is no longer valid}$$

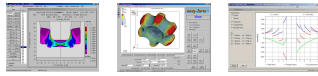
- as new state value the dislocation density  $\rho$  is introduced like

$$\sigma = f(\delta\epsilon/\delta t, T, \rho)$$

- for the dislocation density an evolution equation to describe it's development during the forming operation can be given as

$$\delta\rho/\delta t = (\delta\rho/\delta t)^+ + (\delta\rho/\delta t)^-$$

with  $(\delta\rho/\delta t)^+$ : Processes that generate dislocations  
and with  $(\delta\rho/\delta t)^-$ : Processes that reduce dislocations



## The complex model: "Dislocation Based Model"

### Model idea (2):

- Processes that change the dislocation density

a) dislocation generation

$$\delta\rho/\delta t = M * \delta\varepsilon/\delta t / (b * L_{eff}) \quad \text{mit } L_{eff} = f(\rho)$$

b) dislocation reduction by spontaneous annihilation

$$\delta\rho/\delta t = c_3 * M * d_{spontan} * \delta\varepsilon/\delta t / b * \rho$$

c) dislocation reduction by thermally activated climbing

$$\delta\rho/\delta t = c_4 * (v_k/d_{dipol}) * \rho \quad \text{mit } v_k = f(\sigma) \quad \text{und } d_{dipol} = f(\sigma, \rho)$$

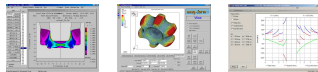
- the yield stress can be calculated as depending of the dislocation density like

$$\sigma_{VM} = M * (\tau_{eff} + \tau_{pass})$$

$$\text{with } \tau_{pass} = \alpha * b * G * \rho^{1/2},$$

$$\text{with } \tau_{eff} = k_B * T / V * \operatorname{asinh}(\arg)$$

$$\text{and with } \arg = (\delta\gamma/\delta t) / (\rho_m * b * \lambda * v) * \exp(Q/k_B * T)$$



## The complex model: "Dislocation Based Model" \*)

### Model idea (3):

- the additional changes of the microstructure can be described adequate equations for

a) dynamic recrystallization,

$$x_{dyn.rek} = f(R_{dyn.rek}, N_{dyn.rek})$$

b) static recovery and static/metadynamic recrystallization,

$$x_{stat.rek} = f(R_{stat.rek}, N_{stat.rek})$$

c) precipitation during hold time and

$$F_p = f(R_p, N_p)$$

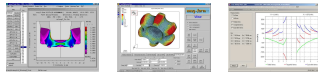
d) grain growth after complete recrystallization

$$R_{gr} = f(R_{dyn.rek}, \delta R_{dyn.rek}/\delta t)$$



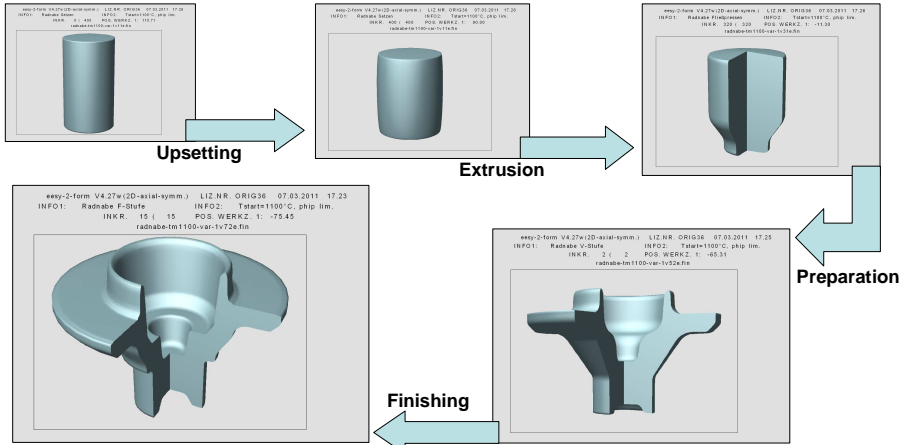
**Dislocation Based Material and Grain model**

\*) Modelling along F. Roters, MPIE Düsseldorf und L. Mosecker, U. Prah, W. Bleck, IEHK RWTH Aachen

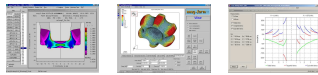


### The complex model: "Dislocation Based Model"

Example: Wheel Hub, 4-step forming at  $T_{start} = 1100^{\circ}\text{C}$ , initial grain size:  $54,8 \mu\text{m}$

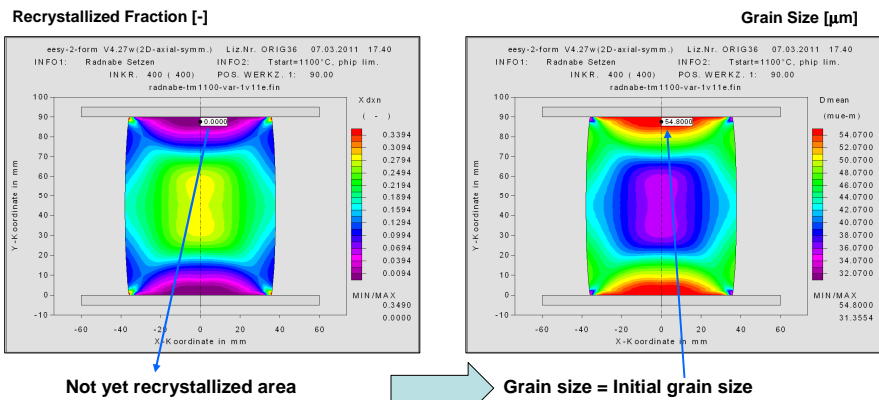


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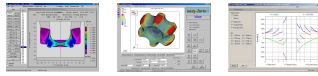


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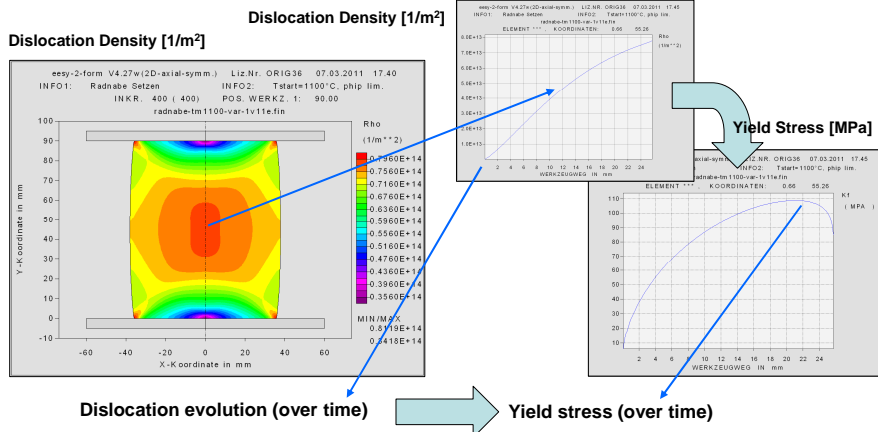


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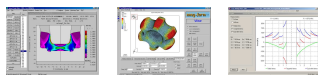


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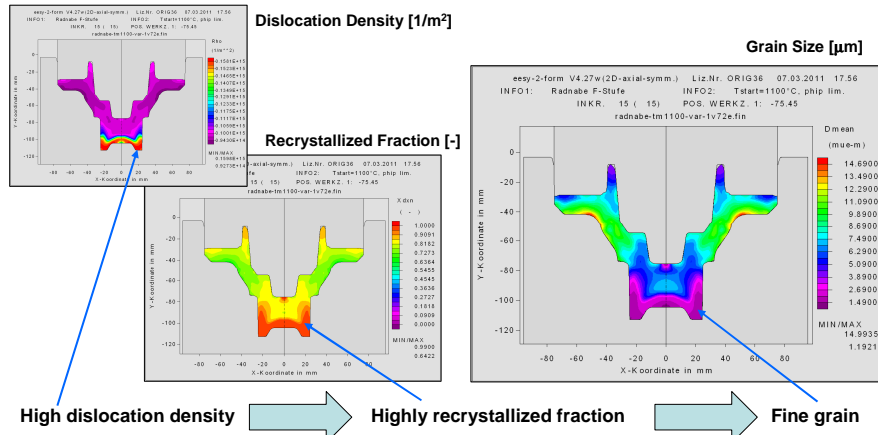


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### The complex model: "Dislocation Based Model"

Example: Wheel Hub, 4-step forming at  $T_{start} = 1100^{\circ}\text{C}$ , initial grain size:  $54,8 \mu\text{m}$



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## *The complex model: "Dislocation Based Model"*

### Results:

- a material model that describes the dependency of microstructure, yield stress and deformation
- yield stress: **calculated based on the dislocation density** and further grain structure parameters as a value independent from a measured curve
- state of recrystallization: **available, split in dynamic und static fraction**
- grain size and grain size distribution: **available as a general value and in its dynamic and static fractions as well**
- Indication of grain growth: **available, quantitative value**
- further microstructure spec. results: **available: i.e. amount and size of precipitations**
- sustainable effects on the material model used: **YES, the yield stress – strain curve will be determined at any time and location out of the dislocation density**



### *Acknowledgement:*

The information for some of the forging parts presented were provide by **Leitritz Turbinenkomponenten Remscheid GmbH, Remscheid and Hirschvogel Umformtechnik GmbH, Denklingen.**

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**German "Bundesministerium für Bildung und Forschung"**



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**"Dislocation Based Model"**

as part of the joint research project

**"Mikrostrukturbasierte Modellierung des Umformverhaltens von mikrolegierten Stählen bei mehrstufigen Schmiedeprozessen"**.